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Electronic structure of $\text{Cu}_{1-x}\text{Ni}_x\text{Rh}_2\text{S}_4$ and CuRh_2Se_4 : Band-structure calculations, x-ray photoemission, and fluorescence measurements

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The electronic structure of spinel-type $\text{Cu}_{1-x}\text{Ni}_x\text{Rh}_2\text{S}_4$ ($x=0.0, 0.1, 0.3, 0.5, 1.0$) and CuRh_2Se_4 compounds has been studied by means of x-ray photoelectron (XPS) and fluorescent spectroscopy. Cu L_3 , Ni L_3 , S $L_{2,3}$, and Se $M_{2,3}$ x-ray emission spectra (XES) were measured near thresholds at Beamline 8.0 of the Lawrence Berkeley Laboratory's Advanced Light Source. XES measurements of the constituent atoms of these compounds, reduced to the same binding energy scale, are found to be in excellent agreement with XPS valence bands. The calculated XES spectra which include dipole matrix elements show that the partial density of states reproduce experimental spectra quite well. States near the Fermi level (E_F) have strong Rh d and S(Se) p character in all compounds. In NiRh_2S_4 the Ni $3d$ states contribute strongly at E_F , whereas in both Cu compounds the Cu $3d$ bands are only ~ 1 eV wide and centered ~ 2.5 eV below E_F , leaving very little $3d$ character at E_F . The density of states at the Fermi level is less in NiRh_2S_4 than in CuRh_2S_4 . This difference may contribute to the observed decrease, as a function of Ni concentration, in the superconducting transition temperature in $\text{Cu}_{1-x}\text{Ni}_x\text{Rh}_2\text{S}_4$. The density of states of the ordered alloy $\text{Cu}_{0.5}\text{Ni}_{0.5}\text{Rh}_2\text{S}_4$ shows behavior that is more "split-band"-like than "rigid-band"-like.

I. INTRODUCTION

Spinel compounds exhibit an extensive variety of interesting physical properties and have potential technological applications. There are a variety of $3d$ ion-based oxide spinels, while the S and Se counterparts usually contain $4d$ or $5d$ atoms. Several of the compounds are superconductors (LiTi₂O₄, CuRh₂S₄, CuRh₂Se₄, etc.), there are unusual magnetic insulators (e.g., LiMn₂O₄ and Fe₃O₄), and recently, a d -electron-based heavy fermion metal has been discovered (LiV₂O₄).¹ The surprisingly high value of the superconducting critical temperature (11 K) in LiTi₂O₄ has never been understood.² Another spinel compound, CuIr₂S₄, is neither magnetic nor superconducting but displays a rather unusual metal-insulator transition that is not yet understood.³ The ternary sulfo- and selenospinel CuRh₂S₄ and CuRh₂Se₄ have been found to be superconducting at $T_c = 4.70$ and 3.48 K, respectively.⁴⁻¹⁴ They have the typical spinel structure [$Fd\bar{3}m$] where Cu ions occupy the A tetrahedral sites and Rh ions occupy the B octahedral sites.

This wide range of phenomena in the spinel-structure oxide compounds raises very general questions about the elec-

tronic structure of the sulfides and the selenides: are there indications of strong correlations effects, or can their properties be accounted for as Fermi liquids described by conventional band theory? Different models for the valence of Cu in these compounds have been discussed,^{5,6} but according to recent photoemission measurements given for CuV₂S₄,¹⁵ CuIr₂S₄, CuIr₂Se₄,¹⁶ and Cu_{0.5}Fe_{0.5}Cr₂S₄,¹⁷ Cu is best characterized as monovalent in spinel compounds. Therefore, one expects that the Rh ion will have a formal mixed valence of +3.5 in CuRh₂S₄ and CuRh₂Se₄, and indeed both are good metals. However, very little of the typical temperature-dependent behavior of "mixed valence compounds" is seen in these Rh-based spinels.

The electrical and magnetic properties of $\text{Cu}_{1-x}\text{Ni}_x\text{Rh}_2\text{S}_4$ have been presented by Matsumoto *et al.*¹⁸ The superconducting transition temperature decreases (4.70 K \rightarrow 3.7 K \rightarrow 2.8 K \rightarrow <2.0 K) as Cu is replaced by Ni ($x=0.00, 0.02, 0.05, \text{ and } 0.10$), but the reason for this behavior is unexplained. Hagino *et al.*⁴ have presented extensive data on CuRh₂S₄ and CuRh₂Se₄ (resistivity, susceptibility, magnetization, specific heat, NMR), but their differences do not yet have any microscopic interpretation. Only for CuRh₂S₄ have

general (full potential, all electron) band-structure calculations been reported.¹⁹

In this paper, we present x-ray spectroscopic studies of the valence band electronic structure of these materials. To provide a clear interpretation of this data, we also report first-principles band-structure calculations [linear-augmented-plane-wave method (LAPW)] for CuRh_2S_4 , CuRh_2Se_4 , NiRh_2S_4 , and $\text{Cu}_{0.5}\text{Ni}_{0.5}\text{Rh}_2\text{S}_4$ that enable us to address the properties of these spinels. Total and partial densities of states (DOS), plasma energies and transport-related quantities are calculated as well as x-ray emission spectra. The total and partial DOS and calculated x-ray emission spectra are found to compare favorably with the measured x-ray photoelectron spectra (XPS) and x-ray emission spectra (XES) (which probe total and partial DOS, respectively). All spectral measurements are performed using the same samples which were used to study the electrical and magnetic properties of $\text{Cu}_{1-x}\text{Ni}_x\text{Rh}_2\text{S}_4$ in Ref. 18.

II. EXPERIMENTAL DETAILS

Mixtures of high-purity fine powders of Cu, Ni, Rh, S, and Se with nominal stoichiometry were heated in sealed quartz tubes at 850° C for a period of 10 days. Subsequently, the specimens were reground and sintered in pressed parallelepiped form at 850° C for 48 h. X-ray-diffraction data confirms the spinel phase in these powder specimens. The lattice constants of $\text{Cu}_{1-x}\text{Ni}_x\text{Rh}_2\text{S}_4$ are 9.79, 9.79, and 9.71 Å for $x=0.0$, 0.1, and 1.0, respectively, and 10.27 Å for CuRh_2Se_4 .

The XPS measurements were performed with an ESCA spectrometer from Physical Electronics (PHI 5600 ci, with monochromatized Al K_α radiation of a 0.3 eV fullwidth at half maximum). The energy resolution of the analyzer was 1.5% of the pass energy. The estimated energy resolution was less than 0.35 eV for the XPS measurements on the copper and nickel sulfides. The pressure in the vacuum chamber during the measurements was below 5×10^{-9} mbar. Prior to XPS measurements the samples were cleaved in ultrahigh vacuum. All the investigations have been performed at room temperature on the freshly cleaved surface. The XPS spectra were calibrated using an Au foil to obtain photoelectrons from the Au $4f_{7/2}$ subshell. The binding energy for Au $4f_{7/2}$ electrons is 84.0 eV.

X-ray fluorescence spectra were measured at Beamline 8.0 of the Advanced Light Source at Lawrence Berkeley Laboratory. The undulator beam line is equipped with a spherical grating monochromator,²⁰ and an experimental resolving power of $E/\Delta E=300$ was used. The fluorescence end station consists of a Rowland circle grating spectrometer. The Ni L_3 and Cu L_3 XES were measured with an experimental resolution of approximately 0.5–0.6 eV and S $L_{2,3}$ and Se $M_{2,3}$ with resolution of 0.3–0.4 eV. The incident angle of the p -polarized monochromatic beam on the sample was about 15°. The Cu L_3 and Ni L_3 XES were measured just above the L_3 threshold but below the L_2 threshold which prevented overlap of the metal L_3 and metal L_2 spectra.

III. METHOD OF CALCULATION

The band-structure calculations were done with the full potential LAPW code WIEN97.²¹ The sphere radii were cho-

sen as 2.1, 2.2, and 2.0 a.u. for Cu/Ni, Rh, and S/Se, respectively. The plane-wave cutoff was $K_{\text{max}}=3.25$ a.u., resulting in slightly more than 1400 basis functions per primitive cell (~ 100 basis functions/atom). The local-density approximation (LDA) exchange-correlation potential of Perdew and Wang²² was used. Because the Fermi level falls on a peak in the DOS for NiRh_2S_4 , as shown in Fig. 7, the gradient correction to the LDA exchange-correlation potential of Perdew, Burke, and Ernzerhof²³ was used in the DOS calculations shown in Fig. 7. A mesh of 47 k points in the irreducible zone (Blöchl *et al.*'s modified tetrahedron method²⁴) was used in achieving self-consistency.

The XES spectra were calculated using Fermi's golden rule and the matrix elements between the core and valence states (following the formalism of Neckel *et al.*²⁵). The calculated spectra include broadening for the spectrometer and core and valence lifetimes. The DOS calculations used 47 k points (again, Blöchl's modified tetrahedron method was used). The experimental lattice constants (listed in the previous section) were used in the calculations and the values used for the internal parameter u were taken to be 0.385 for all three stoichiometric compounds (CuRh_2Se_4 , CuRh_2S_4 , NiRh_2S_4) as well as for $\text{Cu}_{0.5}\text{Ni}_{0.5}\text{Rh}_2\text{S}_4$. Experimental data for the internal parameter was not available, so the values were taken to be 0.385 (rather than the "ideal" position of 3/8) by analogy to the related CuIr_2S_4 and CuIr_2Se_4 spinel compounds for which the u parameter has been measured.³

IV. DISCUSSION OF SPECTROSCOPIC DATA

A. CuRh_2S_4 and NiRh_2S_4

The calculated total and partial DOS of CuRh_2S_4 and NiRh_2S_4 , shown in Figs. 1 and 2, respectively, reveal many common features. The valence bands extend from E_F (taken as the zero of energy) to approximately -7 eV and the Fermi level lies near the top of a Rh d -chalcogen p complex of bands that lie below a gap centered 0.5–1.0 eV above the Fermi level. The gap between the valence band and conduction band is found to be about 0.5–0.7 eV wide. The sulfur states in CuRh_2S_4 and NiRh_2S_4 show similar DOS, S $3s$ atomic like states in the region $-12.7 \sim -14.7$ eV and band-like S $3p$ states which are mixed with Rh $4d$ and Cu/Ni $3d$ states in a wide energy region. Cu/Ni $3d$ states are found to be much narrower than Rh $4d$ states which are less localized and form several large peaks in the DOS near the bottom and the top of the valence band. Our results for CuRh_2S_4 are similar to those of Ref. 19 except for the distribution of Cu $3d$ DOS.²⁶ As seen in Fig. 1, Cu $3d$ states lie within the region of S $3p$ states but are weakly hybridized, forming a 1 eV wide peak centered around -2.5 eV. The S d character is quite small and probably reflects tails of the neighboring atoms more than atomic $3d$ character.

The total DOS at the Fermi level [$N(E_F)$] increases from NiRh_2S_4 (8.18 states/eV/cell) to CuRh_2S_4 (9.89 states/eV/cell) which has the same trend as electronic specific-heat coefficients measured in Refs. 4 and 27. For the intermediate compound $\text{Cu}_{0.5}\text{Ni}_{0.5}\text{Rh}_2\text{S}_4$, $N(E_F)$ is 8.43 states/eV/cell, much nearer that of NiRh_2S_4 . In CuRh_2S_4 the Cu $3d$ partial DOS is very small at the Fermi level whereas Rh $4d$ and S $3p$ partial DOS are the main contribution to the total. Con-

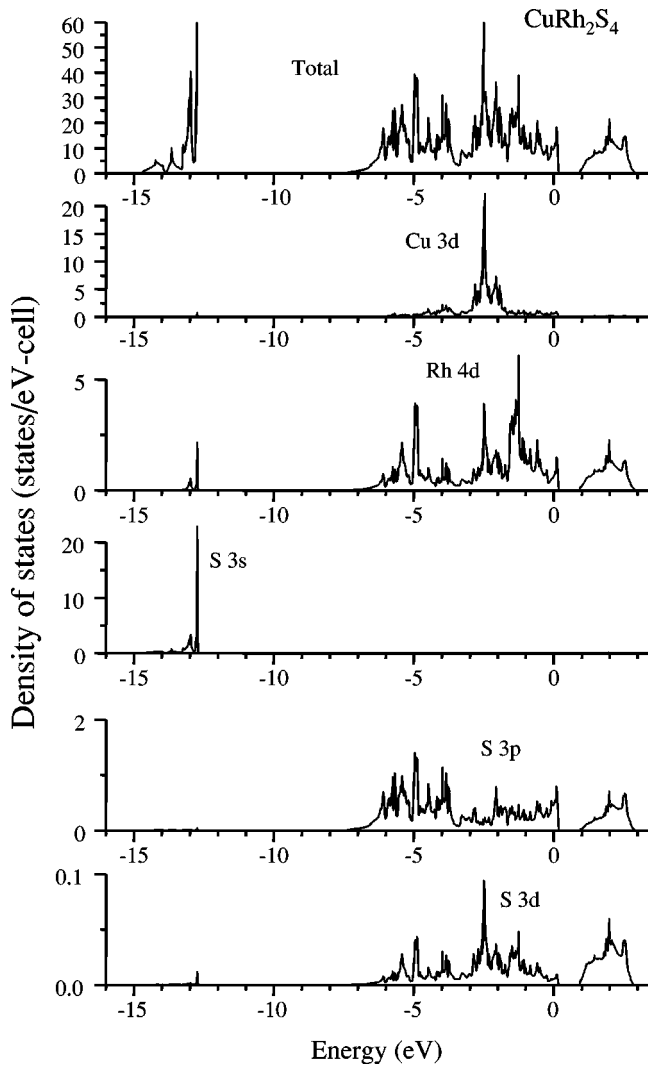


FIG. 1. Calculated total (top panel) and partial DOS in CuRh_2S_4 . Note the hybridization gap that lies just above the Fermi level (taken as the zero of energy), indicating the Fermi level lies in a bonding region of the electronic structure.

sequently, the Cooper pairs in the superconducting state of CuRh_2S_4 are formed mainly by the electrons in the hybridized bands derived from Rh 4d and S 3p states. Several characteristic materials parameters are collected in Table I for easy comparison.

In NiRh_2S_4 the situation is quite different. Ni 3d states are broader and at lower binding energy than the Cu 3d states of CuRh_2S_4 , and hybridization with S p leads to Ni 3d character over a 3 eV wide region that extends above the Fermi level. The result is that the main contribution to the DOS at the Fermi level is from Ni 3d states, unlike in CuRh_2S_4 where the Cu 3d contribution at E_F is very minor.

The experimental Cu L_3 ($3d4s \rightarrow 2p$ transition), Ni L_3 ($3d4s \rightarrow 2p$ transition), and S $L_{2,3}$ ($3s3d \rightarrow 2p$ transition) XES probe Cu 3d4s, Ni 3d4s, and S 3s3d partial DOS in the valence band and, in the first approximation, can be directly compared with calculated band structures. The comparison of the calculated and measured partial DOS are shown in Figs. 3 and 4, where Cu L_3 , Ni L_3 , and S $L_{2,3}$ XES are converted to the binding-energy scale using our XPS measurements of the corresponding core levels

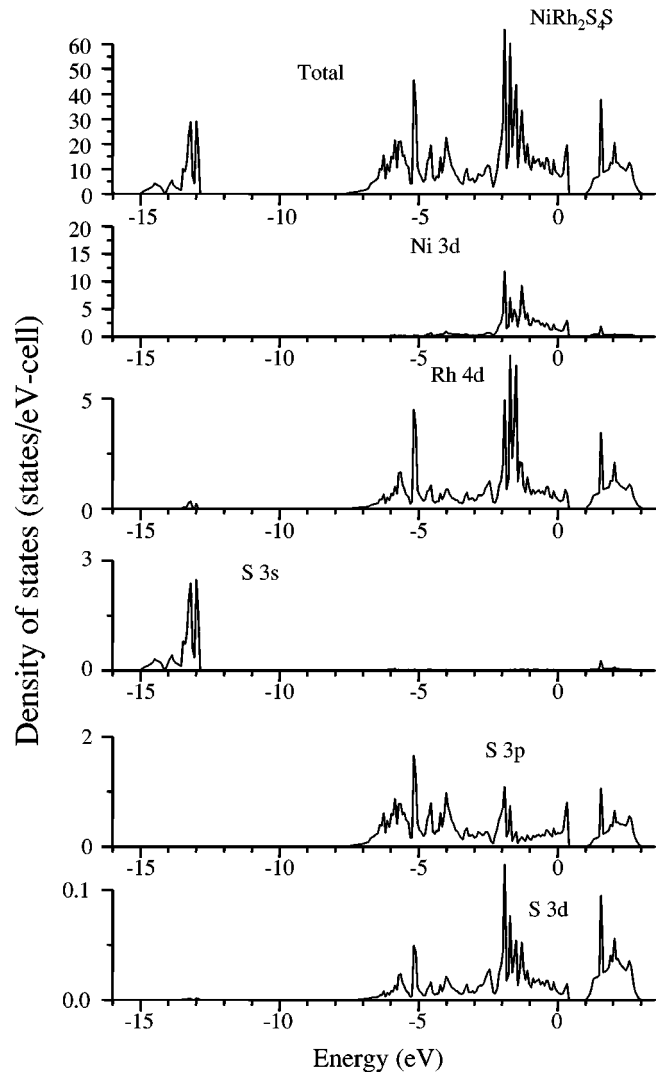


FIG. 2. Calculated total (top panel) and partial DOS as in Fig. 1 but for NiRh_2S_4 .

[$E_{\text{b.e.}}(\text{Cu } 2p) = 932.39$ eV, $E_{\text{b.e.}}(\text{Ni } 2p) = 852.98$ eV and $E_{\text{b.e.}}(\text{S } 2p) = 161.57$ eV]. We see that the measured Cu L_3 , Ni L_3 , and S $L_{2,3}$ XES peaks are very close to Cu 3d, Ni 3d, and S 3s partial DOS in CuRh_2S_4 (Fig. 3) and NiRh_2S_4 (Fig. 4). In each case, the peaks in the calculated DOS lie at somewhat lower binding energy: 1 eV for S 3s and Cu 3p, but only a few tenths of eV for Ni 3d. The difference reflects a self-energy correction that lies beyond our band theoretical methods. In addition, we calculated the emission intensities of Cu/Ni L_3 , Rh N_3 ($4d \rightarrow 4p$ transition)²⁸ and S $L_{2,3}$ XES in both compounds as described in Sec. III. The calculated spectra are presented in the same figures (Figs. 3 and 4) and show close correspondence with experimental spectra as well as with the corresponding partial DOS. From the close agreement, we conclude that the influence of core holes in the measured XES spectra is minor and experimental spectra can be understood directly from the calculated spectra and partial DOS.

B. $\text{Cu}_{1-x}\text{Ni}_x\text{Rh}_2\text{S}_4$ ($x=0, 0.1, 0.3, 0.5, 1.0$)

We measured XPS valence band (VB) spectra for the $\text{Cu}_{1-x}\text{Ni}_x\text{Rh}_2\text{S}_4$ ($x=0, 0.1, 0.3, 0.5, 1.0$) system (see Fig. 5)

TABLE I. Transport related quantities and other data.

	NiRh_2S_4	$\text{Cu}_{0.5}\text{Ni}_{0.5}\text{Rh}_2\text{S}_4$	CuRh_2S_4	CuRh_2Se_4
a (Å)	9.71	9.75 (assumed)	9.79	10.27
$N(E_F)$ (states/eV cell)	8.18	8.43	9.89	12.05
$N(E_F)$ Hagino <i>et al.</i> (Ref. 4)		12.6	13.4	
v_F (10^7 cm/s)	2.49	2.22	1.79	1.75
$\hbar\Omega_p$ (eV)	2.41	2.17	1.89	1.89
T_c (K)	<2.0		4.70	3.483 (Ref. 4)

and found a four-peak structure: (a , c , d , e) for CuRh_2S_4 and (a , b , d , e) for NiRh_2S_4 , each of which is very close to the corresponding calculated total DOS (Figs. 1 and 2). Based on our calculations, we can conclude that the a peak at 1 eV binding energy is formed by Rh $4d$ -S $3p$ states for CuRh_2S_4 and Ni $3d$ -Rh $4d$ -S $3p$ states for NiRh_2S_4 . The next peak (b for NiRh_2S_4 at 2 eV binding energy and c for CuRh_2S_4 at 3 eV binding energy) can be attributed mainly to Ni (respectively Cu) $3d$ states. The d peak (5.5 eV) relates to Rh $4d$ -S $3p$ states and the e peak is associated with atomiclike S $3s$ states. In the solid solution $\text{Cu}_{1-x}\text{Ni}_x\text{Rh}_2\text{S}_4$ the positions of the peaks do not change as the concentration varies, but only the ratio of intensities of b (Ni $3d$) and c (Cu $3d$) peaks vary according to the Cu/Ni concentration.

This behavior suggests that the electronic structure of the solid solution $\text{Cu}_{1-x}\text{Ni}_x\text{Rh}_2\text{S}_4$ can be deduced by analyzing the endpoints ($x=0.0$ and 1.0), CuRh_2S_4 and NiRh_2S_4 . This conclusion results not from a rigid-band picture (which does

not hold) but from the opposite ‘‘split-band’’ behavior²⁹ in which both Cu and Ni retain their own DOS peaks (see Fig. 11) which then vary in strength roughly as the concentration. In Fig. 6 we have compared XPS VB measurements with Cu L_3 , Ni L_3 , and S $L_{2,3}$ XES spectra for $\text{Cu}_{0.5}\text{Ni}_{0.5}\text{Rh}_2\text{S}_4$.³⁰ We see that positions of the peaks in the Ni L_3 , Cu L_3 , and S $L_{2,3}$ XES spectra correspond exactly to peaks b , c , and e of the XPS VB measurements, which is consistent with our interpretation of the XPS data as indicating a solid solution of $\text{Cu}_{1-x}\text{Ni}_x\text{Rh}_2\text{S}_4$ if the split-band behavior holds.

In Fig. 7 we have compared the calculated total DOS of CuRh_2S_4 , NiRh_2S_4 , and $\text{Cu}_{0.5}\text{Ni}_{0.5}\text{Rh}_2\text{S}_4$. With respect to the top of the highest occupied bands, the Fermi energy is highest in the bands of CuRh_2S_4 to accommodate the two additional electrons from the Cu atoms. The behavior of the DOS for the three systems shown are quite different, particularly for Cu and Ni ions, in an energy range between the Fermi levels for NiRh_2S_4 and for CuRh_2S_4 , invalidating a rigid-band interpretation of the differences and similarities in these compounds. This is not surprising given the different

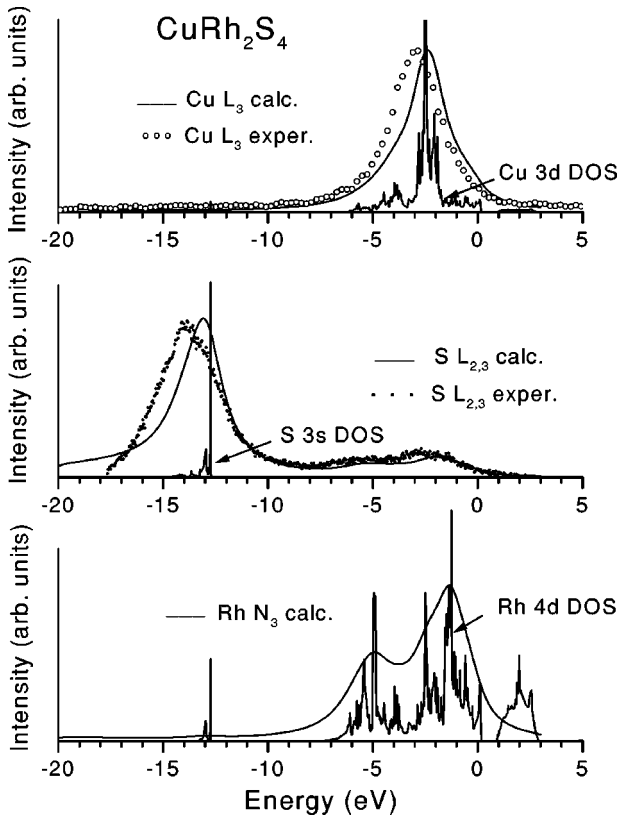


FIG. 3. Comparison of calculated XES and partial DOS with experimental spectra of CuRh_2S_4 . Calculations used the LAPW method as described in the text.

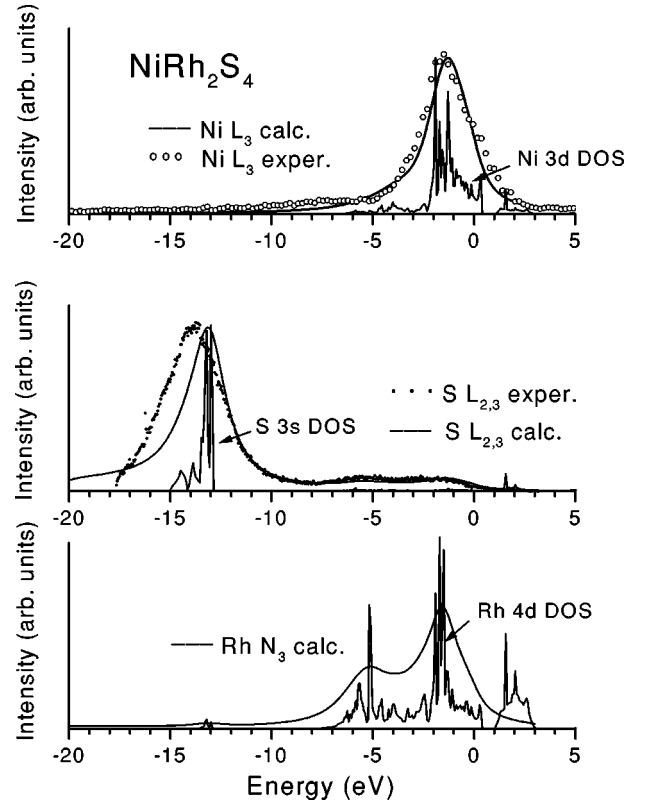


FIG. 4. Comparison of calculated XES and partial DOS with experimental spectra of NiRh_2S_4 .

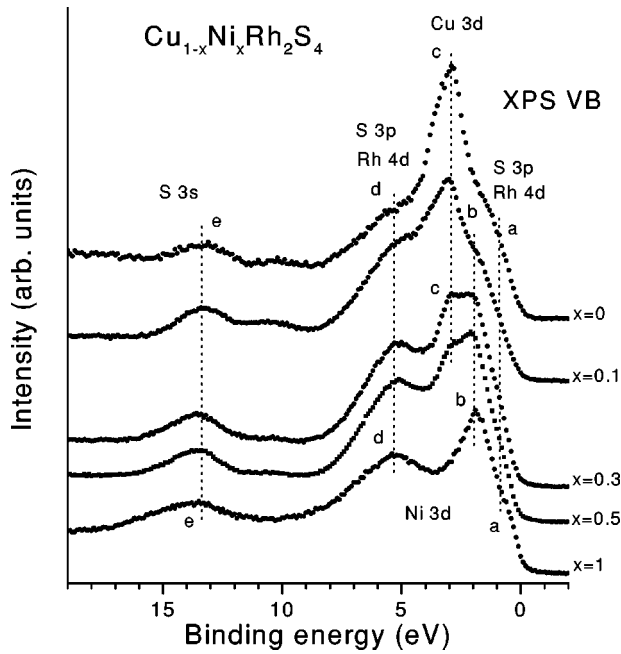


FIG. 5. XPS VB of $\text{Cu}_{1-x}\text{Ni}_x\text{Rh}_2\text{S}_4$ ($x=0.0, 0.1, 0.3, 0.5, 1.0$). The peaks and shoulders *a, b, c, d, e* are discussed in the text.

character of the Ni- and Cu-derived states in this energy region. As mentioned above, whereas states at the Fermi level in NiRh_2S_4 have a strong Ni 3d character, Cu 3d states lie entirely below the Fermi level in CuRh_2S_4 . The character of states at the Fermi level in CuRh_2S_4 are primarily Rh *d*-like states hybridized with S 3*p* states.

According to Ref. 18, the superconducting transition temperature of $\text{Cu}_{1-x}\text{Ni}_x\text{Rh}_2\text{S}_4$ decreases with increasing Ni concentration from 4.7 K ($x=0.0$) to 3.7 K ($x=0.02$) and then to 2.8 K ($x=0.05$). While we attribute this to a general decrease in DOS at the Fermi level as the Ni concentration is increased (see Sec. V), this trend does not require a simple rigid-band interpretation. In the alloy, the DOS within a few tenths of an eV of E_F probably cannot be described by either the rigid band or split-band models.

C. CuRh_2Se_4

Figure 8 shows the calculated total and partial DOS for CuRh_2Se_4 . While it is similar to that of CuRh_2S_4 (Fig. 1), we

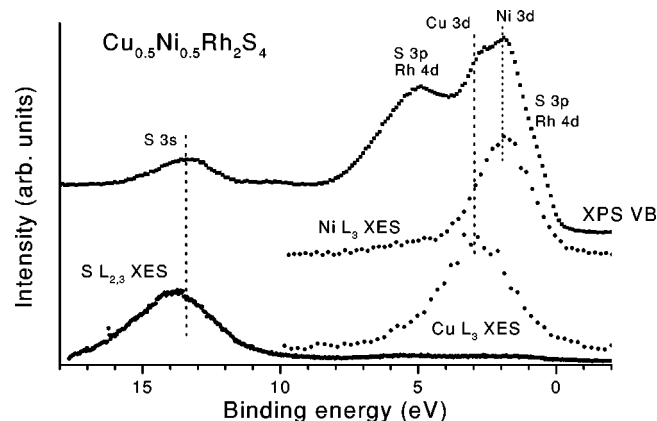


FIG. 6. Comparison of the valence band XPS spectrum (upper set of data) to the Cu L_3 , Ni L_3 , and S $L_{2,3}$ XES in $\text{Cu}_{0.5}\text{Ni}_{0.5}\text{Rh}_2\text{S}_4$. Note the close alignment of XPS and XES peaks.

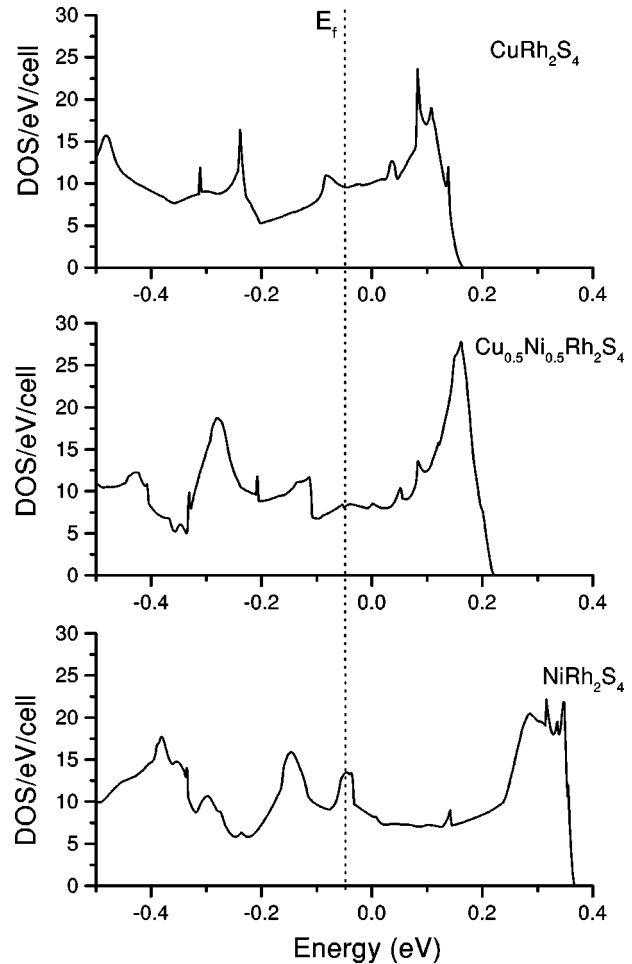


FIG. 7. Calculated total DOS of NiRh_2S_4 , $\text{Cu}_{0.5}\text{Ni}_{0.5}\text{Rh}_2\text{S}_4$, and CuRh_2S_4 aligned to the top of the valence band. Note that, despite the general similarities, a rigid-band-interpretation is not applicable.

can point out two differences: (i) the Se 4*p* DOS is redistributed somewhat compared to S 3*p* and has a higher contribution in the vicinity of the Fermi level, and (ii) the Se *d*-like character is even less than that of the *d*-like character in CuRh_2S_4 . The total DOS at the Fermi level is 12.05 states/eV cell which is higher than in CuRh_2S_4 , in qualitative agreement with measurements of electronic specific-heat measurements.⁴

In Fig. 9 the experimental Cu L_3 and Se $M_{2,3}$ ($4s \rightarrow 3p$ transition) XES measurements are compared to the Cu 3*d* and Se 4*s* partial DOS and calculated spectra. The agreement of the peak positions between experiment and theory is quite close. Again we note that calculated XES spectra exactly follow the partial DOS, as in the case of CuRh_2S_4 and NiRh_2S_4 (Figs. 3 and 4). The XPS valence band data is compared with the Cu L_3 and Se $M_{2,3}$ XES spectra of Fig. 10. The location of Cu 3*d*-Se 4*s*-derived bands is reproduced well (comparable to that in the sulfide) by the calculations. There are some differences in ratio of the XPS peaks for CuRh_2Se_4 and CuRh_2S_4 : the relative intensity of Cu 3*d* peak located at around 2.5 eV is less in CuRh_2Se_4 than in CuRh_2S_4 . This may be due to the 2.5 times larger photoionization cross section of Se 4*p* states as compared to that of S 3*p* states.³¹

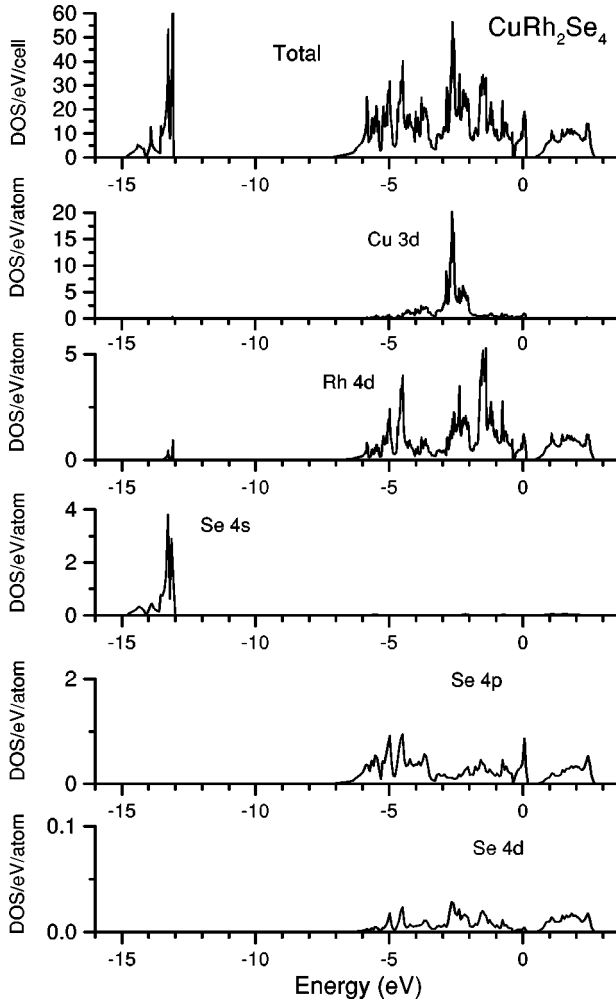


FIG. 8. Calculated total and partial DOS in CuRh_2Se_4 , as shown for CuRh_2S_4 in Fig. 1.

V. OTHER DATA

In a metal the Drude plasma energy tensor $\hbar\Omega_{p,ij}$ contains a good deal of information about low-temperature transport and low-frequency optical properties. $\Omega_{p,ij}$ is given by

$$\Omega_{p,ij}^2 = 4\pi e^2 \frac{1}{V} \sum_{\mathbf{k}} v_{k,i} v_{k,j} \delta(\varepsilon_{\mathbf{k}} - \varepsilon_F) = 4\pi e^2 \langle v_i v_j \rangle N(\varepsilon_F), \quad (1)$$

where $v_{k,i}$ is the i th Cartesian coordinate of the electron velocity, V is the normalization volume, and $\langle \dots \rangle$ indicates a Fermi surface average. The optical conductivity (specializing now to cubic metals) contains a δ -function contribution at zero frequency proportional to Ω_p^2 (which is broadened by scattering processes), and the static conductivity in Bloch-Boltzmann theory³² becomes

$$\rho(T) = \rho_0 + \frac{4\pi}{\Omega_p^2 \tau} \quad (2)$$

(ρ_0 is the residual resistivity at $T=0$) as long as the mean free path $l = v_F \tau$ is large enough that scattering processes are independent. When phonon scattering dominates, which is

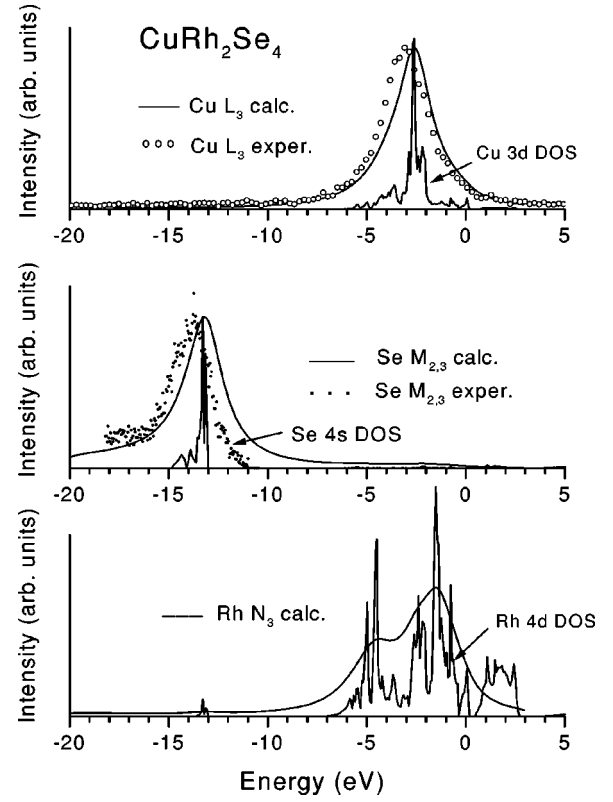


FIG. 9. Comparison of calculated XES and partial DOS with experimental spectra of CuRh_2Se_4 . The agreement of the main features is within 1 eV (Cu and Se) and even better for Rh.

usually the case above 25% of the Debye temperature, the relaxation time τ becomes approximately³³

$$\frac{\hbar}{\tau_{\text{ep}}} = 2\pi \lambda_{\text{tr}} k_B T, \quad (3)$$

where λ_{tr} is a ‘‘transport’’ electron-phonon (EP) coupling strength that is usually close to the EP coupling constant λ that governs superconducting properties. Then in the high- T regime we obtain the estimate

$$\lambda \approx \lambda_{\text{tr}} \approx \frac{\hbar \Omega_p^2}{8\pi^2 k_B} \frac{d\rho}{dT}. \quad (4)$$

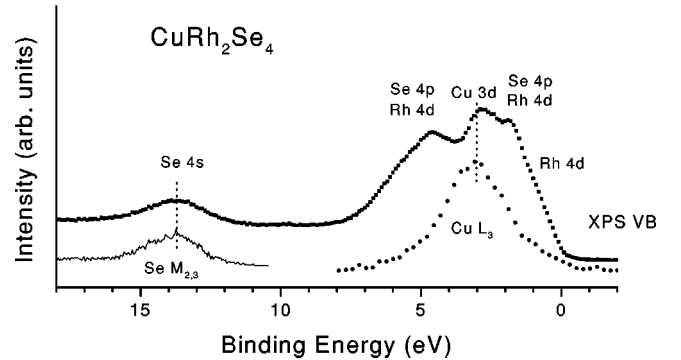


FIG. 10. Comparison of XPS VB to Cu L_3 , and $\text{Se M}_{2,3}$ XES in CuRh_2Se_4 . Note the close alignment of the peaks.

Hagino *et al.* (Ref. 4) have presented resistivity data on sintered samples of CuRh_2S_4 and CuRh_2Se_4 . Although both are clearly metallic ($dp/dT > 0$), the magnitudes of ρ differ by a factor of 20 over most of the range $50 \text{ K} \leq T \leq 300 \text{ K}$. CuRh_2S_4 has $\rho_0 = 2 \mu\Omega \text{ cm}$, indicating excellent metallic behavior in spite of the intergrain scattering that is present in the sintered samples. The CuRh_2S_4 sample had $\rho_0 = 500 \mu\Omega \text{ cm}$ (perhaps from intergrain scattering connected to differences in surface chemistry of the sulfide and the selenide) which makes Eq. (2) inapplicable. Moreover, both materials (especially CuRh_2S_4) show saturation behavior which makes the Bloch-Boltzmann analysis less definitive. However, we can apply this formalism to CuRh_2Se_4 to obtain an estimate, using $dp/dT \approx 2 \mu\Omega \text{ cm/K}$ to obtain $\lambda_{\text{tr}} = 1.8$. This value is almost a factor of 3 larger than $\lambda = 0.64$ found by Hagino *et al.* to be sufficient to account for $T_c = 3.5 \text{ K}$. We expect that the magnitude of ρ measured on the sintered sample of CuRh_2Se_4 , although small, is still not representative of the bulk.

From their measurements, Hagino *et al.*⁴ inferred almost indistinguishable values of the linear specific-heat coefficient γ , the density of states $N(E_F)$, and electron-phonon coupling strengths λ for CuRh_2S_4 and CuRh_2Se_4 . (See Table I.) Our calculations lead to a 20% higher value of $N(E_F)$ in the selenide which is at odds with their values. The 1.2 K lower value of T_c in the selenide is not very definitive, since this difference could be related to softer phonon frequencies. The nearly factor of 2 increase in the susceptibility in the selenide (and not in the sulfide) below 300 K remains unexplained. Data on single-crystal samples may be necessary to resolve these discrepancies.

VI. CONCLUSIONS

The main results of the present study of the electronic structure in $\text{Cu}_{1-x}\text{Ni}_x\text{Rh}_2\text{S}_4$ and CuRh_2Se_4 can be summarized as follows. The electronic states near E_F consist mainly of Rh 4d and S(Se) 3p(4p) orbitals for CuRh_2S_4 and CuRh_2Se_4 and primarily Ni 3d with some Rh 4d and S 3p orbitals in NiRh_2S_4 . Thus, we find that the character of the states at the Fermi level changes in a non-rigid-band way in $\text{Cu}_{1-x}\text{Ni}_x\text{Rh}_2\text{S}_4$, and while there is a general trend of a decreasing DOS at the Fermi level as a function of Ni concentration, we have found that the superconducting trends in $\text{Cu}_{1-x}\text{Ni}_x\text{Rh}_2\text{S}_4$ cannot be explained quantitatively by the calculated DOS of the $\text{Cu}_{1-x}\text{Ni}_x\text{Rh}_2\text{S}_4$ system. Moreover, such an interpretation would be at odds with the partial DOS which shows the different character of states near E_F . The measured x-ray data suggests interpreting $\text{Cu}_{1-x}\text{Ni}_x\text{Rh}_2\text{S}_4$ as a solid state solution more in line with a ‘‘split-band’’ interpretation. The calculated partial DOS for the 50-50 alloy, see Fig. 11, also suggests this interpretation.

Calculated x-ray emission spectra are found to be in an excellent agreement with experimental data, with peak positions differing by only 0.3–1.0 eV. This agreement implies that core hole effects are negligible. In addition to total DOS, plasma energies have been calculated and used to offer additional theoretical input (see Table I) to interpret the differences between CuRh_2S_4 and CuRh_2Se_4 . Unfortunately, transport data appears to be too strongly affected by intergrain scattering to allow a quantitative analysis.

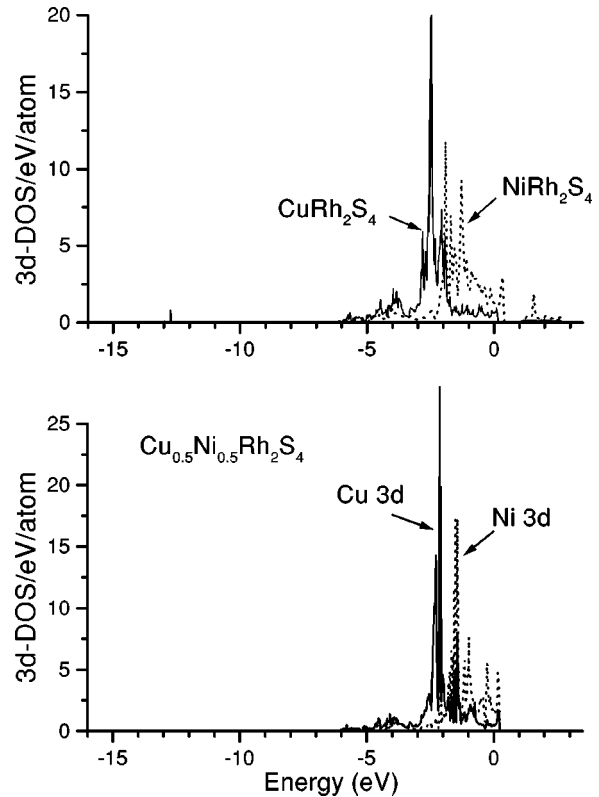


FIG. 11. Comparison of d bands from Ni and Cu in CuRh_2S_4 and NiRh_2S_4 vs $\text{Cu}_{0.5}\text{Ni}_{0.5}\text{Rh}_2\text{S}_4$. The significantly different DOS profiles of Ni and Cu d states in the pure phases discounts a rigid-band interpretation. In $\text{Cu}_{0.5}\text{Ni}_{0.5}\text{Rh}_2\text{S}_4$ we see that the Cu and Ni d bands do not mix very strongly, supporting a ‘‘split-band’’ interpretation.

To summarize, the very good agreement between the measured and calculated electronic spectra indicate a lack of any strong correlation effects. The decrease in superconducting T_c with Ni concentration is likely due to a decrease in $N(E_F)$. Beyond these general conclusions, however, several questions remain. The linear specific heat coefficients are not accounted for quantitatively; neither are the intermediate temperature resistivities, but these must be measured on single crystals to obtain a good experimental picture. Finally, the temperature dependence of the susceptibility of CuRh_2S_4 remains unexplained.

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